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**APPLICATION FOR LETTERS PATENT**

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THERMOPLASTIC POLYURETHANES

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## **THERMOPLASTIC POLYURETHANES**

### **RELATED APPLICATION**

**[0001]** This application claims the priority filing date of U.S. Provisional Application Serial No. 60/451,703 filed March 4, 2003.

### **FIELD OF THE INVENTION**

**[0002]** Desired combinations of physical properties such as slow annealing, relatively low sensitivity of the complex viscosity to changes in shear rates, reduced crystallinity, and good hydrolytic resistance are imparted to thermoplastic polyurethanes when small amounts of a copolyether and a co-chain extender are utilized.

### **BACKGROUND OF THE INVENTION**

**[0003]** U.S. Patent 6,140,453, assigned to Merquinsa Mercados Quimicos, S.L. of Montmelo, Spain relates to specific polyurethane polymers of the formulation set forth therein.

**[0004]** European Patent Application 0 953 586, assigned to Merquinsa Mercados Quimicos, S.L. of Montmelo, Spain relates to specific polyurethane polymers of the formulation set forth therein.

**[0005]** U.S. Patent 4,245,081, assigned to Bayer Aktiengesellschaft of Leverkusen, Germany relates to a continuous process for the production of thermoplastic polyurethanes by reacting A. one or more substantially linear polyols having molecular weights in the range from 400 to 10,000, B. one or more organic diisocyanates and C. a hydroxyl-group-containing chain extender having a molecular weight below 250, the ratio of the NCO-groups in component (B) to the Zerewitinoff-active groups in components (A) and (C) lying between 0.90 and 1.2, in extruders, characterized in that the chain extender (C) used is a mixture of at least two different glycols, of which one is present in a quantity of from 50 to 99% by weight, based on the total amount of component (C), and a second in a quantity of from 1 to 50% by weight, based on the total amount of component (C).

**[0006]** U.S. Patent 4,371,684, assigned to Bayer Aktiengesellschaft of Leverkusen, Germany relates to a process extruding or melt roll calendering thermoplastic

polyurethanes wherein the polyurethanes used are synthesized from p1 (a) one or more relatively high molecular weight substantially linear polyols having molecular weights in the range from 400 to 10,000 (b) diphenyl methane and/or hexamethylene diisocyanate and (c) a mixture of 85 to 99% by weight of 1,4-butane diol and 1 to 15% by weight of at least one co-extender selected from the group consisting of ethylene glycol, diethylene glycol, triethylene glycol, diethylene 1,2-propane diol, 1,3-butane diol, 1,6-hexane diol, 2-ethyl-1,3-hexane diol, 2,2-dimethyl-1,3-propane diol, 1,4-bis-hydroxymethyl cyclohexane, hydroquinone-bis-hydroxyethyl ether, and mixtures thereof, the equivalent ratio of NCO groups in component (B) to the Zerewitinoff-active hydrogen atoms in components (A) and (C) being from 0.9:1 to 1.2:1 and the molar ratio of component (A) to (C) being from 1:20 to 5:1.

[0007] U.S. Patent 6,022,939, assigned to Bayer Aktiengesellschaft of Leverkusen, Germany relates to A thermoplastic polyurethane elastomer is disclosed obtained by reacting A) diisocyanates, B) polyhydroxy compounds and/or polyamines, with C) as chain extenders mixtures of C1) benzene substituted with at least two hydroxyalkyl, hydroxyalkoxy, aminoalkyl and/or aminoalkoxy groups and C2) an alkanediol with 4 to 44 C atoms. The reaction is further characterized in that the molar ratio C1 to C2=60 to 40 to 95 to 5 and in that the equivalent ratio of NCO groups to the sum of the NCO-reactive groups is about 0.9 to 1.20.

### SUMMARY OF THE INVENTION

[0008] Thermoplastic polyurethanes are made utilizing a mixture of polyester and polyether polyols, polyisocyanates which are predominately diisocyanates, and primary chain extenders and a co-chain extender. Small amounts of the polyether polyol are utilized to improve hydrolytic resistance and improve rheological characteristics and small amounts of an asymmetric or a different length co-chain extender are utilized to reduce crystallinity, reduce the sensitivity to annealing, improve rheological characteristics and hydrolytic resistance. It is an important aspect of the present invention to utilize specific ratios of the polyether copolyol to the polyester polyol, specific molar ratios of the co-chain extender to the primary chain extender and specific ratios of the co-chain extender to the copolyol to achieve suitable properties.

### DETAILED DESCRIPTION OF THE INVENTION

[0009] The thermoplastic polyurethanes of the present invention are generally made by combining and reacting a) at least one polyester-polyol component having at least one and desirably two hydroxyl end groups, b) at least one polyisocyanate component which preferably is a diisocyanate and, c) at least one primary or symmetric chain extender, optionally, but preferably with a catalyst. It is an important aspect of the present invention to utilize a specific range of polyether copolyol with the polyester polyol as well as to utilize a specific range of a co-chain extender with the symmetric chain extender. The thermoplastic polyurethane are preferentially made in a twin screw extruder.

### POLYOLS

[0010] The thermoplastic polyurethanes of the present invention comprise hydroxyl terminated polyester polyols. These polyesters are generally linear and have number average molecular weight,  $M_n$ , typically in the range of about 500 to about 5,000, desirably from about 600 to about 4,000, and preferably from about 700 to about 2,500. The number average molecular weight can be determined, for example, by assay of the number of terminal functional groups for a given weight of polymer. Suitable hydroxyl terminated polyesters generally have an acid number of about 1.3 or less and typically about 0.8 or less. The acid number refers to the number of milligrams of potassium hydroxide needed to neutralize one gram of the hydroxyl terminated polyester.

[0011] As known to the art and to the literature, the hydroxyl terminated polyester polymers can be produced by either 1) an esterification reaction of one or more dicarboxylic acids, or anhydrides using one or more glycols or, 2) an esterification reaction of one or more esters of dicarboxylic acids with one or more glycols. An excess mole ratio of glycol to acid or anhydride is utilized so as to achieve a preponderance of terminal hydroxyl groups.

[0012] Suitable dicarboxylic acids for preparing a hydroxyl terminated polyester intermediate can be aliphatic, cycloaliphatic, aromatic or combinations thereof. A single dicarboxylic acid or a combination of dicarboxylic acids can be used. Typically, the dicarboxylic acids have a total of from 4 to about 15 carbon atoms. Examples of suitable

dicarboxylic acids include succinic, glutaric, adipic, pimelic, suberic, azelaic, sebacic, dodecanedioic, isophthalic, terephthalic, and cyclohexane dicarboxylic acids, and the like. Anhydrides of the above dicarboxylic acids, such as phthalic anhydride, tetrahydrophthalic anhydride, and the like, can also be used. Preferred acids include adipic, suberic, sebacic and azelaic.

**[0013]** If the transesterification route for formation of the hydroxyl terminated polyester is utilized, esters of the dicarboxylic acids described above can be used. These esters typically include an alkyl group, usually having 1 to 6 carbon atoms, in place of the acidic hydrogen of the corresponding acid functionalities.

**[0014]** The glycols which are reacted to form the hydroxyl terminated polyester intermediate can be aliphatic, aromatic, or combinations thereof. The glycols typically have a total of from 2 to 12 carbon atoms. Suitable glycols include, for example, ethylene glycol, 1,2-propanediol, 1,3-propanediol, 1,3-butanediol, 1,4-butanediol, 1,5-pentanediol, 1,6-hexanediol, 2,2-dimethyl-1,3-propanediol, 1,4-cyclohexanedimethanol, decamethylene glycol, dodecamethylene glycol, and the like. Preferred glycols include 1,4-butanediol and 1,6-hexanediol.

**[0015]** The hydroxyl terminated polyester intermediates or polyols of the present invention are commercially available from Crompton Corp as Fomrez<sup>®</sup>, Inolex as Lexorez<sup>®</sup>, or from Polyurethane Specialties as Millester<sup>®</sup>.

**[0016]** In order to improve properties such as hydrolytic resistance and rheological characteristics such as reduced sensitivity of the complex viscosity to angular frequency or shear rate, it is an important aspect of the present invention is to utilize a copolyol of at least one polyether polyol in small amounts such as from about 1 to about 50, desirably from about 2 to about 25, and preferably from about 5 to about 20 parts by weight per 100 parts by weight of total amount polyol, i.e., amount of the one or more polyether polyols and that of the one or more polyester polyols.

**[0017]** In order to achieve the desired properties for the thermoplastic polyurethane, only specific types of copolyols are utilized, i.e. hydroxyl terminated polyether polyols. The polyether intermediates or polyols are derived from a diol or polyol having from 2 to about 15 carbon atoms and preferably from 2 to about 6 carbon atoms. The hydroxyl

terminated polyether intermediates can be formed from the reaction of an alkyl diol or glycol with an ether, such as an alkylene oxide having from 2 to about 6 carbon atoms.

[0018] The polyether intermediate can be either a homopolymer or a copolymer. For example, a hydroxyl terminated polyether copolymer polyol can be produced by first reacting propylene glycol with propylene oxide followed by a subsequent reaction with ethylene oxide. Examples of suitable polyether polyols include, but are not limited to, poly(ethylene oxide) which can be formed by reacting ethylene oxide with ethylene glycol; poly(propylene oxide), which can be formed by reacting propylene oxide with ethylene glycol; poly(propylene oxide-co-ethylene oxide), which can be formed by reacting propylene oxide and ethylene oxide with ethylene glycol; and poly(tetramethylene ether glycol) derived from tetrahydrofuran.

[0019] The polyether copolyols of the present invention have number average molecular weights generally from about 100 to about 10,000, desirably from about 250 to about 5,000, and preferably from about 500 to about 4,000, as determined by assay of the number of terminal functional groups for a given weight of polymer. Blends of various polyols can be utilized in the present invention. Suitable polyether polyols are commercially available from Bayer Corporation as Arcol<sup>®</sup>, Acclaim<sup>®</sup> or Multranol<sup>®</sup>; Dupont as Terathane<sup>®</sup>; Arch as Poly G<sup>®</sup>; and The BASF Corporation as PolyTHF<sup>®</sup>. Preferred polyether polyols include poly(ethylene oxide), poly(propylene oxide) and poly(ethylene oxide-co-polypropylene oxide), with poly(tetramethylene ether glycol) (PTMEG) being especially preferred.

[0020] In lieu of or in addition to either the polyester polyol or the polyether polyol or both, a copolymer containing ester groups as well as ether groups, as set forth herein above, can be utilized. A preferred example of such a polyol is poly(butylene adipate)-co-poly(tetramethylene ether). The proportions of the repeat groups therein are generally in the same weight amounts as set forth hereinabove with regard to the polyester and polyether polyols.

### POLYISOCYANATE

[0021] The polyisocyanates of the present invention generally have the formula  $R(NCO)_n$ , where n is usually an integer of 2 to 4 with about 2 being preferred. R can be

an aromatic, cycloaliphatic, an aliphatic, or combinations thereof having from 2 to about 20 carbon atoms. Examples of polyisocyanates include, but are not limited to, diphenylmethane-4,4'-diisocyanate (MDI); toluene-2,4-diisocyanate (TDI); toluene-2,6-diisocyanate (TDI); methylene *bis* (4-cyclohexylisocyanate (H<sub>12</sub> MDI); 3-isocyanatomethyl-3,5,5-trimethyl-cyclohexyl isocyanate (IPDI); 1,6-hexane diisocyanate (HDI); naphthalene-1,5-diisocyanate (NDI); 1,3- and 1,4-phenylenediisocyanate; triphenylmethane-4,4',4''-triisocyanate; polyphenylpolymethylenepolyisocyanate (PMDI); m-xylene diisocyanate (XDI); 1,4-cyclohexyl diisocyanate (CHDI); isophorone diisocyanate; isomers and mixtures or combinations thereof. The preferred isocyanates are MDI and H<sub>12</sub>MDI.

### CHAIN EXTENDERS

**[0022]** The primary or majority chain extenders of the present invention are desirably symmetric, that is possess 2 and preferably 3 elements of symmetry, including, for example, rotation reflection and rotation inversion axes, such as unbranched, unsubstituted straight chain alkane diols free of heteroatoms other than in the functional groups with zerewitinoff hydrogens, e.g., oxygens in the hydroxyl groups, certain cycloaliphatic diols, or certain alkylaryl diols. Examples include 1,6-hexanediol, 1,3-propanediol, 1,5-pentanediol, 1,4-butanediol, 1,4-cyclohexanedimethanol (CHDM), hydroquinone di(β-hydroxyethyl) ether (HQEE), and 1,4-benzenedimethylol. 1,4-butanediol (1,4-BDO) is preferred.

**[0023]** In order to obtain improved properties such as better hydrolytic resistance and reduced tendency to anneal, it is desirable to use small amounts of a co-chain extender which desirably is asymmetric, has a different chain length or is non-linear, in order to reduce crystallinity of the thermoplastic polyurethane. Acyclic chain extenders that contain one or more heteroatom other than in the functional groups with zerewitinoff hydrogens, e.g., oxygens in the hydroxyl groups, are also desirably used as co-chain extenders. Co-chain extenders thus include 1,3-butanediol, neopentyl glycol, diethylene glycol, dipropylene glycol, di(β-hydroxyethyl) resorcinol and 1,2-propylene glycol with 1,3-butanediol (1,3-BDO) and dipropylene glycol (DPG) being preferred. The total amount of the one or more co-chain extenders is generally from about 1 to about 50,

desirably from about 2 to about 25, and preferably from about 5 to about 10 moles per every 100 moles of the primary chain extender such as 1,4-butanediol.

### MOLE RATIO OF COMPONENTS

**[0024]** The overall total mole ratio of the one or more diisocyanates to all of the various dihydroxyl terminated compounds, that is the one or more polyester polyols, the one or more polyether copolyols, the primary chain extender, and the one or more co-chain extenders is from about 0.95 to 1.05 and desirably from about 0.98 to about 1.03. The mole ratio of the chain extenders, that is the primary chain extender and the one or more co-chain extenders, to the polyols, that is the one or more polyester polyols and the one or more polyether polyols, is generally from about 0.4 to about 10, desirably from about 0.6 to about 5, and preferably from about 0.7 to about 3. The mole ratio of the co-chain extender to the wt % of the polyether copolyol is generally from about 0.1 to about 10, desirably from about 0.15 to about 3, and preferably from about 0.2 to about 2. This mole ratio of co-chain extender to the wt.% of polyether polyol is important inasmuch as it imparts lower crystallinity.

### CATALYSTS

**[0025]** Generally any conventional thermoplastic polyurethane catalyst known to the literature and to the art can be utilized in preparing the thermoplastic polyurethane of the present invention. Such catalysts include organic and inorganic acid salts of, and organometallic derivatives of, bismuth, tin, iron, antimony, cobalt, thorium, aluminum, zinc, nickel, cerium, molybdenum, vanadium, copper, manganese and zirconium, as well as phosphines and tertiary organic amines. Representative organotin catalysts have from about 6 to about 20 carbon atoms and include stannous octoate, dibutyltin dioctoate, dibutyltin diluarate, and the like. Representative tertiary organic amine catalysts include triethylamine, triethylenediamine, N,N,N',N'-tetramethylethylenediamine, N,N,N',N'-tetraethylethylenediamine, N-methylmorpholine, N-ethylmorpholine, N,N,N',N'-tetramethylguanidine, N,N,N',N'-tetramethyl-1,3-butanediamine, N,N-dimethylethanolamine, N,N-diethylethanolamine, and the like. Representative



polyalcohol amine catalysts include triethanolamine, diethanolamine, or bis(2-hydroxyethyl)amino-2-propanol, and the like.

**[0026]** The amount of catalyst employed is generally less than about 1000 and desirably less than about 400 parts by weight per million parts by weight of the total weight of the polyisocyanate(s), the polyol components, and the chain extenders. Mixtures of the above noted catalysts can likewise be utilized. It is desirable to use minimal amounts of the catalyst in order to minimize side reactions. Preferred catalysts include stannous octoate, dibutyltin dioctoate, dibutyltin dilaurate, and bismuth octoate.

### **ADDITIVES**

**[0027]** In addition to the above-identified components, the polyurethane compositions of the present invention can also contain various additives, fillers, pigments, or dyes, and the like, utilized in conventional amounts which are well known to the art and to the literature. Generally additives are utilized which impart desired properties to the thermoplastic polyurethanes such as various antioxidants, various ultraviolet light inhibitors, waxes such as amide waxes and ester waxes, thickening agents, and the like. The fillers, when utilized, are generally mineral fillers, that is inorganic, and include ground mica, talc, kaolin clay, calcium carbonate, calcium sulfite, colloidal silica, fumed silica, wollastonite, hollow glass microspheres, glass, carbon and graphite fibers, various metallic oxides such as zinc, titanium zirconium, and the like, ground quartz, various metallic silicates, metallic powders such as lead, aluminum, bronze, and the like.

**[0028]** If it is desired that the thermoplastic polyurethane composition of the present invention have a color or hue, any conventional pigment or dye can be utilized in conventional amounts. Hence, any pigment known to the art and to the literature can be utilized as for example titanium dioxide, iron oxide, carbon black, and the like, as well as various dyes provided that they do not interfere with the various urethane reactions.

**[0029]** The polyurethanes of the present invention have A Shore hardness, as measured according to ASTM D-2240, generally about 98A or less, desirably from about 70A to about 98A, and preferably from about 80A to about 98A.

### REACTION PROCEDURES

**[0030]** The thermoplastic polyurethanes of the present invention can be prepared by various methods known to the art and the literature. For example, a two-step process can be utilized wherein the polyester polyol and the polyether copolyol are reacted with at least one diisocyanate to form a prepolymer which is subsequently chain extended with the above- noted chain extenders. A preferred process is a one-shot procedure wherein generally all the reactants are brought together and substantially simultaneously reacted. A highly preferred procedure is a random melt polymerization process wherein the polyester polyol, the polyether copolyol, the linear chain extender, and the different length chain extender as well as the catalyst are brought together and mixed at a temperature of from about 60°C to about 100°C. The blend is then heated to a temperature of from about 110°C to about 200°C, and preferably from 120°C to 180°C. The diisocyanate such as MDI is heated to temperatures in the same range as the blend, for example about 120°C, and then mixed with the blend. The reactants are mixed thoroughly during the reaction period which is generally from about 2 to 3 minutes. The reaction is exothermic and thus exhibits an increase of temperature generally greater than 70°C. A suitable reaction vessel is a twin screw extruder. Upon completion of the reaction, the polymers are discharged into a cold vessel and are allowed to cool to room temperature. Various physical tests of the thermoplastic polyurethane were then conducted.

**[0031]** The thermoplastic polyurethanes of the present invention can have a weight average molecular weight (Mw) of from about 10,000 to about 1,000,000, desirably from about 30,000 to about 250,000, and preferably from about 60,000 to about 120,000 as measured by gel permeation chromatography against a polystyrene standard.

**[0032]** The thermoplastic polyurethanes of the present invention as compared to simple polyester-based thermoplastic polyurethanes, where the polyurethane block is prepared from a single primary or symmetrical chain extender, generally have reduced crystallinity, reduced tendency to anneal, improved hydrolytic resistance or stability and lower sensitivity to shear.

**[0033]** Accordingly, the polyurethanes can be used to coat fabrics, or other substrates such as adhesive laminates or coatings. Suitable fabrics can be either woven or non-woven such as polyester fibers, polyolefin fibers, nylon fibers, and the like. Industrial

applications include coated films, sheets, or fabrics as for conveyer belts, collapsible storage bags (e.g., fuel, water, fruit juices, food oils, heating oils etc), inflatables (e.g., escape slides and platforms, floatation devices, air-mattresses, life jackets, white-water or life rafts, oil booms, petro-seals, power lifting devices, weather balloons) or grape press membranes, and the like. In the apparel industry, uses include labels and stickers used in laundry and professional outfits, as well as protective clothing/apparel, protective covers, rainwear, sealable coatings for labels, surgical drapes, protective apparel, synthetic leather, tents, upholstery, wet or diving suits, and the like. Other uses include liners for pipe repair, load space covers, and the like.

**[0034]** The polyurethanes can also be used to make unsupported TPU film and sheet via extrusion or calendering. Applications for such films and sheets include air mattresses, shower curtains, aeration sheets for water purification plants, adhesives, equipment covers, protective wear, aprons, body bags, tank liners, pipe liners, and the like.

**[0035]** The following examples serve to illustrate but not to limit the present invention.

**[0036]** The amount of co-chain extender and copolyol (PTMEG with  $M_n = 1000$ ) listed in Table 1 were added to 25g of 1,4-BDO and mixed with an amount of poly(butylene adipate) (PBAd) such that the total weight of said mixture was 200g. The mixture was melted at 60°C and stirred thoroughly. The mixture was stirred and heated to 120°C. Hot MDI (60°C) in an amount consistent with the stoichiometry indicated in the table was added. The mixture was stirred for 3 minutes and then poured into a cool Teflon-coated pan. The material was then aged for 2 hour at 105°C. The invention is depicted by Examples 2, 3, 5, 9, 10, 12-14 and 16 while Examples 1, 4, 6-8, 11 and 15 are comparative examples or controls.

**[0037]** The above recipes were tested with regard to the following physical and rheometrical properties listed in Table 2.

**TABLE 1**  
Formulations on which the invention and comparative examples are based

	Example	PBAD	1,4-BDO (g)	1,3-BDO (g)	DPG (g)	PTMEG (g)	Co-chain extender (mole %)	Copolyol (wt %)	Mole % co- chain extender to copolyol (PEt)	Melt Index (dg/min)	MDI Stoichiometry (%)	Mw
Control	1	172.5	25	2.5	0	0	10	0	-	48	97.8	81500
Invention	2	155.25	25	2.5	0	17.25	10	10	1.0	50	98.8	78000
Invention	3	138	25	2.5	0	34.5	10	20	0.5	38	98.8	94000
Control	4	173.75	25	1.25	0	0	5	0	-	54	97.8	74800
Invention	5	156.37	25	1.25	0	17.38	5	10	0.5	33	97.7	102000
Control	6	157.5	25	0	0	17.5	0	10	0	44	98.4	82200
Control	7	140	25	0	0	35	0	20	0	50	98.3	84300
Control	8	171.28	25	0	3.72	0	10	0	-	45	98.8	81200
Invention	9	154.15	25	0	3.72	17.13	10	10	1.0	70	98.3	69500
Invention	10	137.02	25	0	3.72	34.26	10	20	0.5	64	98.3	75800
Control	11	173.14	25	0	1.86	0	5	0	-	48	97.8	75800
Invention	12	155.83	25	0	1.86	17.31	5	10	0.50	24	99.4	100000
Invention	13	150.4	25	2.1	0	22.5	8.4	13	0.65	34	98.4	101000
Invention	14	139	25	1.25	0	34.75	5	20	0.25	39	98.3	87000
Control	15	175	25	0	0	0	0	0	-	40	98.3	150000
Invention	16	150.4	25	2.1	0	22.5	8.4	13	0.65	45	98.1	79000

TABLE 2  
Physical properties and rheological characteristics of invention and illustrative examples are based

	Example	T <sub>c</sub> (°C)	T <sub>CN</sub>	T <sub>g</sub> (°C)	Original tensile property		Aged tensile property <sup>a</sup>		V <sub>t</sub> (T <sub>m</sub> +15)	V <sub>m</sub> ((T <sub>m</sub> +15)/(T <sub>m</sub> +35))	V <sub>t</sub> (T <sub>m</sub> +35)	V <sub>t</sub>
					TS (psi)	TE (%)	TS <sub>N</sub>	TE <sub>N</sub>				
Control	1	76	0.90	-14	4996	527	0.26	0.38	15	17	7.3	1.2
Invention	2	67	0.80	-9	5454	547	0.37	0.73	4	6	2.4	1.8
Invention	3	67	0.80	-9	5610	545	0.55	0.92	6	7	3.2	1.4
Control	4	78	0.93	-17	4616	559	0.27	0.29	12	26	4.1	4.5
Invention	5	67	0.80	-10	5539	587	0.46	0.89	4	6	2.2	2.9
Control	6	76	0.90	-17	5561	590	0.27	0.58	6	9	3.0	4.9
Control	7	75	0.89	-20	5522	604	0.39	0.79	5	8	2.2	4.7
Control	8	84	1.00	-12	5548	516	0.50	1.03	6	10	3.4	2.4
Invention	9	75	0.89	-14	4487	554	0.39	0.76	4	8	2.2	2.1
Invention	10	74	0.88	-12	5117	584	0.49	0.86	5	7	2.3	3.3
Control	11	77	0.92	-15	5269	565	0.23	0.39	6	10	2.2	4.4
Invention	12	74	0.88	-11	5674	534	0.36	0.86	6	9	3.6	3.1
Invention	13	76	0.90	-16	-	-	-	-	6	8	3.5	1.5
Invention	14	73	0.87	-12	-	-	-	-	5	8	3.1	2.2
Control	15	84	1.00	-15	-	-	-	-	7	13	3.3	4.8
Invention	16	79	0.94	-15	-	-	-	-	5	9	2.4	3.1

a) aged for 4 weeks at 75°C and 95% relative humidity

A dash "-" means that no data was obtained

[0038] Tensile properties, specifically tensile strength, TS, and ultimate elongation, TE, were measured according to ASTM D-412/D-638.

[0039] Reduced crystallinity is determined from the change in the crystallization temperature listed in Table 2 as  $T_C$  as measured by DSC when the sample is heated to 250°C and then cooled at 10°C/min as compared to a control containing a similar primary chain extender content and optionally one or more polyester polyols, i.e. a standard or conventional thermoplastic polyurethane not containing any polyether polyol or co-chain extender, i.e., Example 15. Reduced crystallinity is thus defined as the  $T_C$  normalized to that of said standard TPU,  $T_{CN}$ . According to the invention the values of  $T_{CN}$  must be generally about 0.95 or less, desirably about 0.90 or less, and preferably about 0.85 or less.

[0040] Hydrolytic stability is demonstrated by comparing the tensile strength exhibited by the samples following aging for 4 weeks at 75°C and 95% relative humidity to that exhibited prior to aging or normalized tensile strength,  $TS_N$ . According to the invention the values of this parameter must be greater than about 0.3, and preferably greater than about 0.35.

[0041] The melt index (MI) was measured at 190°C and 8.7 kg, Mw was measured by dissolving the samples in THF and injecting the solution into a GPC apparatus, using polystyrene of known Mw as standards.

[0042] Rheological measurements were performed on a Physica UDS 200 Universal Dynamic Spectrometer System using a MP306 (25mm,0°) HT spindle. The spindle and plate are heated to 190°C. The pre-dried sample (1.6 g) is placed at the mid range of the heated plate and the distance between top of the plate and bottom of the spindle is adjusted to 3 mm and left in that position for 1 minute. The distance between the spindle and plate is then readjusted to 2 mm and again held in this position for 1 minute.

[0043] The gap is then decreased to between top of the bottom plate and bottom of the spindle is 1 mm which is used as is the measuring gap.

[0044] Prior to beginning measurements, the system is left at this gap and temperature for 1 minute, making sure that the gap is filled with sample. Superfluous product is removed from the bottom plate taking care that the outer surface of the

cone is not covered with sample. Time, temperature and frequency dependent measurements are run according to the profiles summarized in the Tables 3 and 4.

**[0045]** The temperatures for rheological measurements were selected by comparison to the melting temperature,  $T_m$ , a conventional TPU of similar composition but without the co-chain extender nor co-polyol. For the examples given and based on Example 15,  $T_m = 155^\circ\text{C}$ .

**[0046]** The reduced sensitivity to shear is determined by monitoring the frequency dependence of the viscosity ( $V_f$ ) as calculated by dividing the complex viscosity ratio at  $6\text{ s}^{-1}$  to that observed at  $600\text{ s}^{-1}$  measured at a temperature equal to  $15^\circ\text{C}$  above the  $T_m$  of the conventional TPU,  $170^\circ\text{C}$ ,  $V_f(T_m+15)$  and  $35^\circ\text{C}$  above the  $T_m$ ,  $190^\circ\text{C}$ ,  $V_f(T_m+35)$ . Suitable  $V_f(T_m+15)$  values are about 10 or less and preferably about 6 or less whereas suitable  $V_f(T_m+35)$  values are desirably about 5 or less and preferably about 4 or less.

**[0047]** The sensitivity of the complex viscosity to temperature is determined by taking the ratio of the complex viscosity measures at  $170^\circ\text{C}$  and  $6\text{ s}^{-1}$  to that measured at  $190^\circ\text{C}$  and  $600\text{ s}^{-1}$ , i.e.  $V_n((T_m+15)/(T_m+35))$ . According to the invention, the values of this parameter must be desirably about 10 or less and preferably about 9 or less.

**[0048]** Reduced annealing is determined by the time dependent viscosity ratio, final/initial, ( $V_t$ ) when the sample was left at  $T_m+15$ ,  $170^\circ\text{C}$ , for 30 minutes. According to the invention, the values of this parameter must be about 4.0 or less, and preferably about 3.5 or less.

**[0049]** To be in compliance with the present invention, generally a majority, desirably at least 5 and preferably all 6 of the above numerical criteria must be met.

**Table 3**  
**Frequency and temperature sweep measurement**

Interval	Program settings	Description
1	Temperature: constant 190°C Interval duration: 1 minute	Sample tempering.
2	Temperature: constant 190°C Amplitude: constant 5% Frequency: ramp log 100 -1 Hz (reverse profile) Number of data points: 40	Frequency sweep measurement of the sample at 190°C.
3	Temperature: ramp lin 190°C – 180°C Cooling rate: 10°C/min Interval duration: 1 min	Cooling.
4	Temperature: constant 180°C Interval duration: 1 minute	Sample tempering.
5	Temperature: constant 180°C Amplitude: constant 5% Frequency: ramp log 100 -1 Hz (reverse profile) Number of data points: 40	Frequency sweep measurement of the sample at 180°C.
6	Temperature: ramp lin 180°C – 170°C Cooling rate: 10°C/min Interval duration: 1 min	Cooling
7	Temperature: constant 170°C Interval duration: 1 minute	Sample tempering.
8	Temperature: constant 170°C Amplitude: constant 5% Frequency: ramp log 100 -1 Hz (reverse profile) Number of data points: 40	Frequency sweep measurement of the sample at 170°C.
9	Temperature: ramp lin 170°C - 190°C Heating rate: 10°C/min Interval duration: 2 min	Switch over from frequency sweep to temperature sweep measurement.
10	Temperature: constant 190°C Interval duration: 1 minute	Sample tempering.
11	Temperature: ramp lin 190°C – 140°C Cooling rate: 10°C/min Interval duration: 5 min Amplitude: constant 5% Frequency: constant 50 Hz Number of data points: 40	Temperature sweep measurement.



**Table 4**  
**Time sweep measurement**

Interval	Program settings	Description
1	Temperature: ramp lin 190°C - 200°C Heating rate: 10°C/min Interval duration: 1 minute	Heating.
2	Temperature: constant 200°C Interval duration: 5 minutes	Sample tempering.
3	Temperature: ramp lin 200°C – 170°C Cooling rate: 10°C/min Interval duration: 3 min	Cooling.
4	Temperature: constant 170°C Interval duration: 1 minute	Sample tempering.
5	Temperature: constant 170°C Amplitude: constant 5% Frequency: constant 10 Hz Interval duration: 30 min Number of data points: 40	Time sweep measurement of the sample at 170°C.
6	Temperature: ramp lin 170°C – 200°C Heating rate: 10°C/min Interval duration: 3 min	Heating.
7	Temperature: constant 200°C Interval duration: 5 minutes	Sample tempering.
8	Temperature: ramp lin 200°C – 170°C Cooling rate: 10°C/min Interval duration: 3 min	Cooling.
9	Temperature: constant 170°C Interval duration: 1 minute	Sample tempering.
10	Temperature: constant 170°C Amplitude: constant 5% Frequency: constant 10 Hz Interval duration: 30 min Number of data points: 40	Repeat of time sweep measurement at 170°C.

**[0050]** As apparent from Table 2, Examples 2, 3, 5, 9, 10, 12-14, and 16 obtained good values with regard to reduced sensitivity to shear viscosity at 170°C and 190°C, i.e.  $V_f(T_m+15)$  and  $V_f(T_m+15)$ , good reduced sensitivity of complex viscosity,  $V_{rf}((T_m+15)/(T_m+35))$ , good reduced annealing,  $V_t$ , and the like.

**[0051]** While in accordance with the patent statutes, the best mode and preferred embodiment have been set forth, the scope of the invention is not limited thereto, but rather by the scope of the attached claims.